

Silver Modified TiO₂/Halloysite Thin Films for Decontamination of Target Pollutants

Dionisios Panagiotaras, Elias Stathatos, Dimitrios Papoulis

Abstract—Sol-gel method has been used to fabricate nanocomposite films on glass substrates composed halloysite clay mineral and nanocrystalline TiO₂. The methodology for the synthesis involves a simple chemistry method utilized nonionic surfactant molecule as pore directing agent along with the acetic acid-based sol-gel route with the absence of water molecules. The thermal treatment of composite films at 450°C ensures elimination of organic material and lead to the formation of TiO₂ nanoparticles onto the surface of the halloysite nanotubes. Microscopy techniques and porosimetry methods used in order to delineate the structural characteristics of the materials. The nanocomposite films produced have no cracks and active anatase crystal phase with small crystallite size were deposited on halloysite nanotubes. The photocatalytic properties for the new materials were examined for the decomposition of the Basic Blue 41 azo dye in solution. These, nanotechnology based composite films show high efficiency for dye's discoloration in spite of different halloysite quantities and small amount of halloysite/TiO₂ catalyst immobilized onto glass substrates. Moreover, we examined the modification of the halloysite/TiO₂ films with silver particles in order to improve the photocatalytic properties of the films. Indeed, the presence of silver nanoparticles enhances the discoloration rate of the Basic Blue 41 compared to the efficiencies obtained for unmodified films.

Keywords—Clay mineral, nanotubular Halloysite, Photocatalysis, Titanium Dioxide, Silver modification.

I. INTRODUCTION

ONE of the main environmental problems which receives special attention in industry is the wastewater treatment. Textile colored effluents, paper pulp and other related industries have influenced the quality of the natural environment because of the wastewater disposal. Azo dyes used in the textile industry are resistant water pollutants provoke toxic effects in the aquatic systems. Azo dyes exhibit low degradation rates in the existing environmental conditions to the ultraviolet and solar light irradiation and also to biological treatment defied the development of innovative treatment technologies [1], [2]. These methods, called Advanced Oxidation Technologies (AOTs) have also used to the decontamination of polluted water. AOTs have a high efficiency to remove azo dyes from water based on the oxidative power of radical species created during these

processes. In addition photocatalysis one of the AOTs methods can be successfully applied to the oxidation and final removal of azo dyes with the formation of carbon dioxide as a final product. A variety of wide band gap semiconductors under UV or solar light have been used for the photocatalytic decomposition of various organic pollutants [3], [4]. Among them, TiO₂ is a high efficient material for the photo-degradation of many organic pollutants, while it is relatively inexpensive, non-toxicity and stable in aquatic environments [5], [6]. Moreover, the mesoporous nanocrystalline anatase TiO₂ particles, films or membranes have extended uses in environmental remediation techniques [7]. The photocatalytic potency of the ultrafine TiO₂ powders is due to the high particle surface area since reactions take place on the surface of the nanocatalyst, but the agglomeration of the ultrafine TiO₂ particles to larger aggregates decay the efficiency of the photocatalyst.

Nevertheless, although TiO₂ used as mobilized catalyst for its high catalytic surface area and activity [8], TiO₂ powders cannot easily be recovered from solutions when they are used for water purification. Highly dispersed ultrafine TiO₂ particles in suspension are difficult to recover from solution during water and wastewater treatment. To overcome this limitation, many researchers immobilize TiO₂ catalyst onto substrates as thin films and membranes. This technique, although lower the catalytic surface area compared to powder, extent the use and application of the photocatalyst because of the easy way to handle these materials in a variety of environmental applications [9]-[11]. A number of factors controlling the photocatalytic activity of the films such as the number of -OH surface groups, the specific surface area, particle morphology, possible aggregation and phase composition. However, modifications on the porous structure of this class of materials increase the photocatalytic properties of the catalysts. Activated carbon, glass slides and fibers, membranes, and zeolites are used among other sustainers in order to enhance the photocatalytic activity of the nanocomposites [12]-[14]. Degrease of the photocatalyst efficiency is due to the lower total surface area and the immobilization of the catalyst compare to the ultrafine pure TiO₂ powder. Nevertheless, much research has been done to overcome this limitation by using highly porous materials such as clay minerals as substrates for TiO₂ particles immobilization [15], [16]. The clay mineral halloysite with tubular structure has been considered as a cheap and appropriate material used for TiO₂ particles immobilization [17], [18]. Additionally, many studies have been devoted to the improvement of titania photoactivity by depositing noble

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metals. In particular, deposition of silver on TiO₂ colloidal particles has received an extended interest that goes beyond photocatalytic degradation. Thus metallic silver can be photocatalytically deposited on TiO₂ and this process can be used in silver recovery e.g. from waste photographic effluents.

In this work we use the sol-gel method with surfactant molecules in order to produce thin mesoporous nanocrystalline TiO₂ films in the presence of clay mineral halloysite nanotubes. We examined the efficiency of the new photocatalyst for the degradation of the azo-dye Basic Blue 41. Furthermore, the as-prepared nanocomposite thin films were modified with silver in order to enhance their photocatalytic efficiency. The silver modification of halloysite/TiO₂ nanocomposite photocatalysts immobilized on glass substrates is a promising technique for the degradation of azo dyes in solution and can be used for the water and wastewater treatment.

II. MATERIALS AND METHODOLOGY

A. Chemical Reagents

Pure and well crystalline halloysite samples with tubular morphology were originated from Utah, USA and they were size fractionated by gravity sedimentation to obtain sizes less than 2 μm. Separation of the clay fraction was carried out by using centrifugation methods. The clay fractions of the most halloysite-rich samples were used for the preparation of TiO₂-halloysite nanocomposites. Commercially available Triton X-100 (X100, polyethylene glycol tert-octylphenyl ether), titanium tetraisopropoxide (TTIP), acetic acid (AcOH), Basic Blue 41 (BB-41), silver nitrate (>99%) and all solvents were purchased from Sigma-Aldrich. Double distilled water with resistivity 18.2 MΩ (Millipore) was used in all experiments.

B. Sol-Gel Synthesis of Composite Halloysite/TiO₂ Sol

The X100 as a nonionic long chain surfactant organic molecule was selected as a pore directing agent in a sol. Such amphiphilic molecules may succeed the existence of ordered mesophase and the ability to adjust large inorganic clusters in aqueous condition at the same time. A suitable amount of X100 was homogeneously dissolved in ethanol. Before adding alkoxide precursor, AcOH was added into the solution for the esterification reaction with ethanol. Then, TiO₂ precursor, TTIP was added at a time under vigorous stirring. The molar ratio of the materials was optimized at X100: Ethanol: AcOH: TTIP = 1: 68: 6: 1 in accordance to previous published results [19]. Halloysite (HAL) powder was mixed with previous solution in various quantities following HAL 5%, 20%, 25% and 30% weight ratio compared to TiO₂ mass. After stirring for several minutes, the dispersion was ready to be used on glass slides. Films prepared on glass slides for the four HAL weight ratios will be referred as HT5, HT20, HT25 and HT30 respectively while HT0 represents pure TiO₂ films without the presence of halloysite.

C. Formation of TiO₂ Thin Films and Silver Modification

Borosilicate glass with a size of L75 mm (effective L60) × W25 mm × T1 mm was used as a substrate for fabricating

immobilized HAL/TiO₂ thin films. Before coating, the substrate was thoroughly cleaned with detergent and washed with water and acetone and finally dried in a stream of nitrogen. A home-made dip-coating apparatus equipped with a speed controller to maintain a withdrawal rate of ~10 cm/min was used to dip in and pull out the substrate from the sol. After coating, the films were dried at room temperature for 1 hour, calcined in a multi-segment programmable furnace (PLF 110/30, Protherm) at a ramp rate of 5°C/min to 450°C for 15 min, and cooled down naturally. Only one layer of catalyst was formed for all HAL/TiO₂ ratios. In case that the as resulted films was chosen to be modified with silver, they were immersed in 1mM silver nitrate aqueous solution. The films were remained in the sol for 15 minutes and then they were rinsed with double distilled water and dried under nitrogen gas. Afterwards, they were exposed to black light (UV source) for 20 minutes while their color was turned to brown which means that adsorbed silver ions were reduced and they were converted to zero valence silver.

D. Instrumentation and Materials Characterization

Nitrogen intrusion/extrusion curves were measured with a Micromeritics Tristar 3000 and the surface area, porosity, and pore size distribution were derived by differentiating them according to BET method. The values were obtained from thick films after scratching the material due to the difficulty of sample collection from the thin films. A Bruker D8 Advance diffractometer with CuKα (λ = 1.5406 Å) radiation and Bragg-Brentano geometry was employed for X-ray diffraction (XRD) studies of the halloysite-TiO₂ catalyst. For the visual morphology of HAL/TiO₂ nanostructure, an environmental scanning electron microscope (FESEM, Zeiss SUPRA 35VP) was used and inspect film homogeneity. AFM images were obtained with a Nanoscope III, Digital Instruments, in the tapping mode. Absorption measurements of BB-41 sols were carried out with a Hitachi U-2900 UV-Vis spectrophotometer.

E. Photocatalytic Properties of Composite Films

For the photocatalytic experiments a cylindrical reactor was used which was presented in a previous publication [20]. Dry air was pumped through a gas inlet using a small pump to ensure continuous oxygen supply to the reaction solution while it was simultaneously agitated. Four Black lights with 4 W nominal power were placed around the reactor for UV illumination. The whole construction was covered with a cylindrical aluminum reflector. Cooling was achieved by air flow from below the reactor using a ventilator. The catalyst was in the form of four borosilicate glasses, covered on one side with nanocrystalline HAL%-TiO₂ films. The total surface of the photocatalyst films was approximately 60 cm². The reactor was filled with 75 ml of 2.5×10⁻⁵ M BB-41 aqueous solution. BB-41 is strongly adsorbed on pure or halloysite modified TiO₂ films. For this reason, we stored the solution in the presence of the photocatalyst in the dark for an hour and all of our photocatalytic results were obtained after equilibrium. The photocatalytic discoloration process for the dye was examined by monitoring the absorption maximum of

the BB-41 solution ($\lambda=610$ nm) at various irradiation times. Photocatalytic discoloration rate of BB-41 was calculated by the formula:

$$r = \frac{C_0 - C}{C_0} \quad (1)$$

where C_0 is the initial concentration of BB-41 solution and C is the final concentration after irradiation with UV light. Discoloration efficiency was determined as:

$$\text{efficiency}\% = \frac{C_0 - C}{C_0} 100\% \quad (2)$$

For the repeated use of the photocatalysts, the films were washed with double distilled water and dried at 80°C while no further treatment was followed for the films. The adsorption of BB-41 on HAL-TiO₂ films was examined under dark and after 1 hour presence of the films in dye's aqueous solution.

III. RESULTS AND DISCUSSION

A. Structural Characteristics of Halloysite and TiO₂ Nanocomposite Films

The composite Halloysite/TiO₂ films were prepared on borosilicate glass substrates for different HAL weight proportions as described in experimental section. Samples, abbreviated as HT5, HT20, HT25 and HT30 represent the different weight proportions of halloysite in TiO₂ sol while HT0 is referred to pure TiO₂ films. After calcination at relatively high temperature $\approx 450^\circ\text{C}$ to eliminate organic substances all films were firmly attached on the glass without any cracks because of TiO₂ content. Halloysite was finely dispersed in the films for any proportion was used. First, the crystallinity of the films was examined in order to detect any differences to the crystal structure of halloysite after heating and the crystal phase of resulting nanocomposite TiO₂. The XRD patterns of all films are presented in Fig. 1.

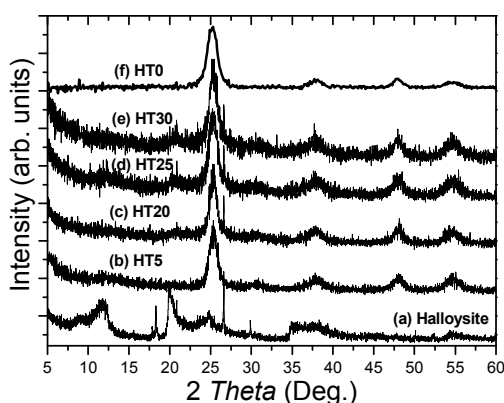


Fig. 1 XRD patterns of Halloysite-TiO₂ nanocrystalline films: (a) halloysite clay mineral (b) HT5, (c) HT20, (d) HT25, (e) HT30, (f) HT0.

Strong reflection at $2\theta=12.2^\circ$ is corresponded to halloysite 7\AA , while a less intensive reflection at 2θ equal to 8.80° corresponds to halloysite 10\AA (Fig. 1, curve a). The main peak of halloysite 7\AA seems that is remaining after heating but it is much lower mainly to the low amount of the mineral in films proving the remaining crystallinity of the clay mineral. A second reason is probably the partial dehydration of halloysite due to the temperature applied for the synthesis of the films. It should be noted that in this sample (Fig. 1, curve a) low amounts of quartz are also present (main reflections observed at 26.60° and 20.80°). Titania pure nanocrystalline film (HT0) is also presented in Fig. 1 (curve f) where a reflection (101) of anatase form at $2\theta=25.1^\circ$ is observed. The two basic reflections at $2\theta=12.2^\circ$ and $2\theta=25.1^\circ$ for HAL and TiO₂ are maintained at the rest of samples with different intensity ratio because of the variable proportion between them. The grain size for TiO₂ has been calculated from XRD patterns using Scherrer's equation:

$$D = 0.9\lambda / (s \cos\theta) \quad (3)$$

where λ is the wavelength of the X-ray and s is the full width (radians) at half maximum (FWHM) of the signal. The crystallite size for TiO₂ is calculated 7.5, 8.1, 11.3, 8.5 and 9.4 nm for samples HT0, HT5, HT20, HT25 and HT30 respectively.

TABLE I
STRUCTURAL CHARACTERISTICS OF HALLOYSITE-TiO₂ FILMS

Sample	Total pore volume V_p (cm ³ /g)	Specific surface area S (m ² /g)	Total porosity ϕ (%)	Mean pore diameter D_{por} (nm)
Halloysite	0.125	50.9	24.46	9.85
HT0	0.133	121.3	33.58	4.78
HT5	0.165	108.6	39.10	6.09
HT20	0.155	77.5	37.62	7.98
HT25	0.171	95.3	39.95	7.20
HT30	0.159	106.2	38.21	6.01

All the peaks indicated that the crystal phase of the materials containing TiO₂ was anatase and the relatively large width of peaks indicated that the size of the nanocrystallite was less than 12 nm. It should be noted that it is evident from the XRD patterns that the calcination at 450°C for 15 min did not damaged halloysite clay.

Because of the difficulty in directly characterizing the porosity of immobilized Halloysite/TiO₂ thin films, the characterizations were carried out on the corresponding particles. The specific surface area S , the total pore volume V_p , the mean pore diameter D_{por} , and the total porosity ϕ were calculated for all samples and they are presented in Table I.

Pure halloysite powder has relatively large pore volume and similar to that obtained for pure TiO₂ while their mixtures possess slightly greater values. As it concerns the particle surface areas, in the case of pure TiO₂ a relatively high value of 121.3 m²/g is measured and 50.9 m²/g for halloysite. All other samples with different proportions of halloysite in TiO₂ matrix appear intermediate values for particle surface areas as they appear in Table I.

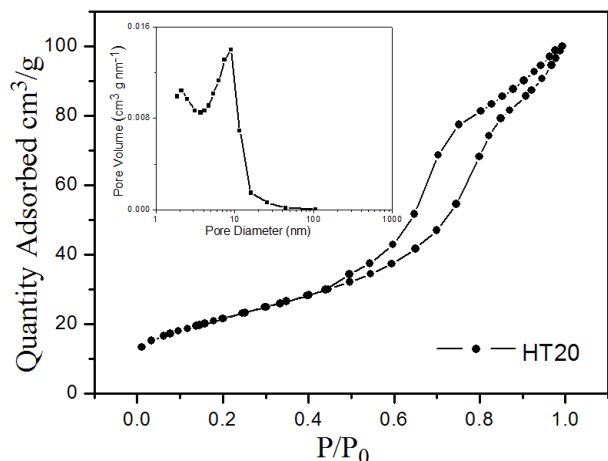


Fig. 2 Sorption-desorption isotherm and pore size distribution (inset) for HT20 thin film

Nitrogen sorption-desorption curves for all Halloysite/TiO₂ samples were obtained. However, as an example we present the data obtained for HT20 films in Fig. 2, while pore size distribution appear as an inset of the same figure. This porosity is, also, apparent in SEM images shown in Fig. 3. Halloysite samples are consisted of tubular particles as it can be seen in Fig. 3 (a). TiO₂–Halloysite modified are presented in Fig. 3 (b). The average diameter of the tubes, as they were observed before modification, is 40-70nm while the length is between 100-500nm. After modification, TiO₂ nanoparticles uniform in size overlay halloysite tubes.

Besides, TiO₂ nanoparticles help to the stabilization of the composite material on the borosilicate glass substrate after calcination by forming stable Ti-O-Si bonds [21], [22]. In Fig. 3 (b) halloysite nanotubes seem to be completely covered with uniform layers of TiO₂ and uniform particle distribution. The thickness of TiO₂ film without halloysite tubular particles is around 180-200 nm with only one dipping layer according to a cross sectional SEM image.

The homogeneity of TiO₂ particles' size and film can be also seen in Fig. 3 (b). According to the same image of Fig. 3 the TiO₂ crystal grains have a spherical shape while they have an average size ranging from 12 to 16 nm. TiO₂ particles were also found to form aggregates on halloysite external surfaces but these were of uniform small size as it was also proved by porosimetry data.

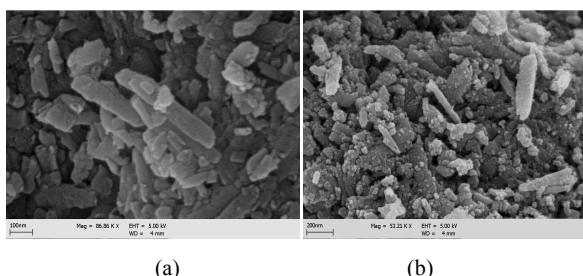


Fig. 3 SEM images of: (a) Halloysite powder, (b) HT30 film

The dispersion of halloysite in TiO₂ films is obvious but it is firmly agglutinated. No cracks or peeling off traces around halloysite boundaries were observed. The film is permanently attached on the glass substrate with good adherence while halloysite cannot be rived from the composite material. The composite films were finally put at a silver nitrate aqueous solutions in order to silver modify the films. The surface of the films after UV exposure for silver zero valance particles formation on TiO₂ is presented to the following AFM image of Fig. 4

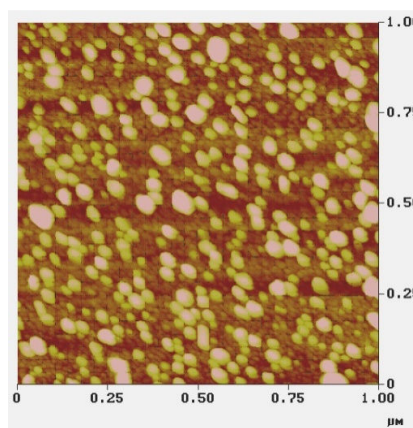


Fig. 4 AFM image of silver modified TiO₂ particles

The evidences of SEM images for the nanocomposite material may help us to schematic represent the film fabrication on glass substrates (Fig. 5). It is assumed that the organophilic interphase, assured by X100 surfactant coating, acts like template medium which provides titanium dioxide nanoparticles with relatively monodispersed particle sizes on the surface. The initially amorphous TiO₂ phase was crystallized after calcination at 450°C for 15 min in air while X100 was completely burned out.

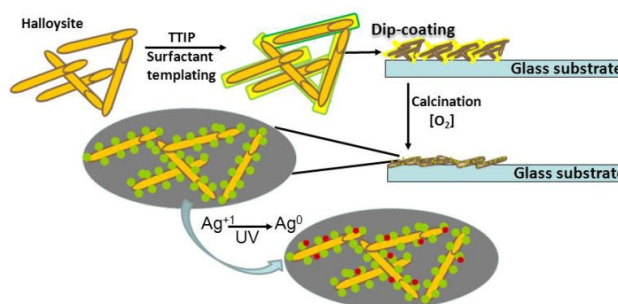
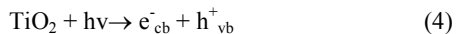


Fig. 5 Preparation procedure for HAL-TiO₂ nanocomposite photocatalyst formation as films

Silanol groups (Si-OH) of clay mineral can react with titanium alkoxide giving covalently bonded organic-inorganic derivatives which could be useful for anchoring metal oxide nanoparticles on halloysite surface.

B. Photocatalytic Properties of HAL-TiO₂ Composite Films

Titanium dioxide mediated photo-degradation involves the generation of electron-hole pairs [5], which migrate to the photocatalyst surface forming surface bound hydroxyl and superoxide radicals according to the following equations:



It is also well known that the hydroxyl and superoxide radicals are the primary oxidizing species in the photocatalytic process. These oxidative reactions result in the photo-discoloration of dyes as target pollutants in water.

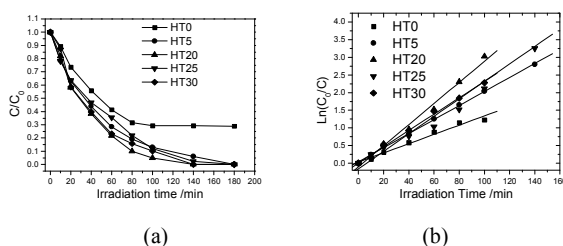


Fig. 6 (a) Photo-discoloration of BB-41 for different halloysite - TiO₂ proportions under UV light, (b) Ln(C₀/C) as a function of irradiation time for HAL-TiO₂ photocatalysts

Photocatalytic experiments were undertaken on Basic Blue 41 to evaluate HAL-TiO₂ composite catalyst in films as shown in Fig. 6. The different weight percentages of halloysite in TiO₂ nanocrystalline films showed variations to the photocatalytic activities of the films. The rate of discolorization was monitored with respect to the change in intensity with time of the absorption peak at 610 nm. The absorption peak of the dye diminished with time and disappeared during the reaction indicating that it had been decomposed. Besides, the UV illumination was started after one hour of the photocatalyst presence in dye's sol in order to be in equilibrium before illumination. The results also show that there was no direct photolysis of BB-41 in the absence of any photocatalyst. In the case of HT20 a complete discolorization was reached within 140 min of illumination implied the synergistic effect between clay mineral and TiO₂ by preparing highly porous HAL-TiO₂ catalysts (Fig. 6 (a)). Moreover, considering the small amount of TiO₂ catalyst immobilized onto the substrate, the HAL-TiO₂ films were highly efficient to degrade the azo dye. As a consequence, the tubular nanocomposite materials can be an alternative substrate for the growth of nanoparticle TiO₂ achieving an efficient photocatalyst. Decomposition kinetics of BB-41 has been observed to follow first-order kinetics and it is well established that photo-discoloration experiments follow Langmuir-Hinshelwood model, where the reaction rate, *r*, is proportional to the surface coverage, *θ*, according to the following equation [23]:

$$r = -\frac{dC}{dt} = k_1\theta = \frac{k_1KC}{1+KC} \quad (7)$$

where *k*₁ is the reaction rate constant, *K* is the adsorption coefficient of the reactant and *C* is the reactant concentration. In the case that *C* is very small, *KC* factor is negligible in respect to unity and the (7) describes first-order kinetics. The integration of (7) yields to the (8):

$$-\ln\left(\frac{C}{C_0}\right) = k_{app}t \quad (8)$$

With limit condition that on *t*=0 we have the initial concentration *C*₀. *k*_{app} is the apparent first-order rate constant. Discoloration kinetics of BB-41 in presence of different HAL proportions in TiO₂ nanocrystalline films is presented in Fig. 6b. The maximum value for rate constant was calculated for sample HT20 (29.8x10⁻³ min⁻¹) while the value for pure TiO₂ film was estimated at 12.9x10⁻³ min⁻¹ (Table II).

TABLE II
CONSTANT OF BB-41 DEGRADATION RATE IN THE PRESENCE OF HALLOYSITE-TiO₂ COMPOSITE FILMS MODIFIED WITH SILVER PARTICLES

Sample	<i>k</i> _{app} (x10 ⁻³ min ⁻¹) with no silver	<i>k</i> _{app} (x10 ⁻³ min ⁻¹) with silver
HT0	12.9	18.5
HT5	19.8	19.8
HT20	29.8	26.6
HT25	24.8	25.3
HT30	23.2	28.5

Furthermore, all the samples HAL-TiO₂ exhibited better performance than pure TiO₂. We mainly attribute this behavior to the internal light scattering because of the presence of the halloysite.

This is also could be attributed to better structural characteristics of the films compared to pure TiO₂ film mainly tabulated at porosity and total pore volume. However, the rate of dye discolorization depends on adsorption of the dye into the catalyst porous structure. Finally, it has been found that the same photocatalysts can be used in several photocatalytic cycles without remarkable loss to their efficiency.

C. Photocatalytic Properties of Silver Modified HAL-TiO₂ Films

As an alternative procedure to further increase the efficiency of the HAL-TiO₂ photocatalysts silver ions were deposited on the TiO₂ surface by submerging TiO₂ films in aqueous solutions of metal salt for several minutes. It generally believed that silver modified TiO₂ could cause a better separation of charge carriers on the oxide surface [24]. Metal cations could be adsorbed onto TiO₂ films at substantial quantities because of the relatively high specific surface area of the films. In the case of silver ions and after their adsorption only one electron is enough to reduce silver ions and create zero valence noble metal on the surface of TiO₂, according to the following equation:



This electron is easy to be generated, either by UV illumination under black light irradiation or by thermal heating of the films [24].

Indeed, the films after UV exposure were turned to light brownish colored attributed to the creation of zero valence silver due to plasmon resonance absorption [25].

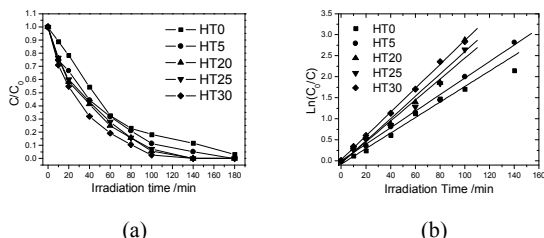


Fig. 7 (a) Photo-discoloration of BB-41 under silver particles modified TiO_2 films in presence of different weight percentages of halloysite irradiated with UV light, (b) $\ln(C_0/C)$ as a function of irradiation time for silver modified HAL- TiO_2 photocatalytic films

The corresponding photocatalytic experiments to evaluate the silver modified HAL- TiO_2 composite catalyst in films are shown in Fig. 7 (a). However, the original salt concentration obviously affects the quantity of deposited metal, affected photo-discoloration efficiency. When the silver salt concentration was $10^{-3} \text{ mol L}^{-1}$ we obtained the highest photocatalytic efficiency.

At higher metal load, efficiency dropped, possibly due to screening effects. Decomposition kinetics of BB-41 in presence of silver modified HAL- TiO_2 films, have been observed to follow first-order kinetics as in the case of bare HAL- TiO_2 films. The data are presented in Table II and Fig. 7 (b).

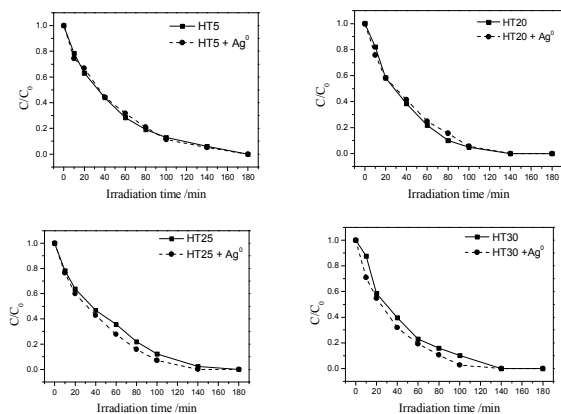


Fig. 8 Comparison between bare and silver modified HAL- TiO_2 films to the photo-discoloration of BB-41

In the case of silver modification the most efficient films are proved to be HT30 where a complete discoloration of the dye was achieved within 100 minutes.

For a direct comparison among HAL- TiO_2 films and silver modified ones to the photocatalytic decomposition of BB-41, we present the data of Fig. 8.

It is obvious that silver modified films exhibit a slight better performance to the discoloration of BB-41 mainly due to the better separation of charge carriers on the oxide surface. Further experiments with several noble metals modified composite HAL- TiO_2 films are under way.

IV. CONCLUSIONS

Nanostructured Halloysite- TiO_2 films were synthesized via sol-gel method composed of ethanol, acetic acid, titanium tetraisopropoxide, halloysite nanotubes and nonionic surfactant molecules as organic template. Slow hydrolysis reaction and stable incorporation of inorganic network onto surfactant molecules made it possible to control the subsequent porous nanostructure. The HAL- TiO_2 films exhibited enhanced structural properties including crystallinity and active anatase phase while enhanced photocatalytic properties to the discoloration of BB-41 in water were succeeded. The experiments on photocatalytic discoloration of BB-41 implied the importance of synergistic effect between clay mineral nanotubes and TiO_2 nanoparticles. In addition, the presence of silver particles on the surface of the composite photocatalyst could further increase the efficiency of the catalyst due to better separation of charge carriers on the TiO_2 .

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